



Letter

Valence state of transition metal center as an activity descriptor for CO₂ reduction on single atom catalysts

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ABSTRACT

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Carbon dioxide (CO₂) catalytic reduction has been passionately pursued for a long period of time due to its special importance in alleviating the greenhouse effect as well as generating valuable fuels and chemicals [1]. Among the various approaches for achieving CO₂ conversion, considerable efforts have been devoted to electrochemical reduction of CO₂ since this technology operates at ambient and mild environments and can potentially produce various useful products [2]. By now, a variety of electrocatalysts based on transition metals, transition metal oxides, and transition metal dichalcogenides [3–5], have been proposed for electrochemical CO₂ reduction reaction (CRR). Unfortunately, the current electrocatalysts are still confronted with two severe bottlenecks: Poor selectivity toward various products in CRR process, and the competition with hydrogen evolution reaction (HER) towards unwanted side-products [6]. The former originates from associated multielectron transfer process of CRR [7], while the latter is mainly due to the fact that the equilibrium potentials for most products of the CRR are very close to HER [8]. Thus, developing highly active and selective catalysts for electrochemical CRR remains one of the challenging issues in this field.

Recently, single-atom catalysts (SACs), where the isolated transition-metal (TM) atoms dispersed on perfect or defective surfaces serve as the active centers, have been widely investigated due

to their superior catalytic performance along with a reduction in the amount of loaded precious metals. The high activity and selectivity can be ascribed to the unique electronic structure of isolated atoms, which is different from their bulk counterparts. SACs have already been investigated in a range of electrocatalysis processes, including oxygen evolution reaction (OER), nitrogen reduction reaction (NRR) and CRR [9–12].

The activity of SACs is significantly affected by the substrate environments. The ideal substrate should be able to anchor single atoms and possess outstanding stability and conductivity. Recently, a family of two-dimensional transition-metal carbides and nitrides (M_{n+1}X_n), known as MXene, were discovered and synthesized in experiments [13]. In the formula, “M” represents early transition metals, X corresponds to C or/and N, and n = 1, 2 or 3. Owing to their high electronic conductivity, large surface areas, and tunable surface composition [14,15], MXenes are experimentally and theoretically proved to be promising candidates for binding metal atoms and for use as catalysts [16–21]. The basal planes of MXenes are usually terminated by functional groups (–OH, –F, and –O) after hydrogen fluoride treatment during their synthesis [13]. Nevertheless, recent studies have confirmed the possible elimination of –F group [13,22] and the –OH groups can be converted into –O groups after high-temperature annealing [23]. Therefore –O terminations are more representative of the real surface. In this work, we have theoretically investigated the potential of single transition metal (TM = Ti, V, Cr, Mn, Fe, Co, Ni, Cu, Zr, Nb,

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Mo, Ru, Rh, Pd, Ag, Hf, Ta, W, Os, Ir, Pt, Au) atoms decorated on Ti_2CO_2 , a typical MXene [24], as electrochemical catalysts for CRR (calculation details can be found in the Supporting Information). Our computational results demonstrate that $\text{Mn@Ti}_2\text{CO}_2$, $\text{Co@Ti}_2\text{CO}_2$ and $\text{Ni@Ti}_2\text{CO}_2$ are ideal candidates for electrochemical CO_2 reduction to CH_4 , CH_3OH and HCOOH , respectively. More importantly, we found that TM-to-substrate electron donation, as varied with different TM atoms, can be used as a simple descriptor to predict catalytic activity of $\text{TM@Ti}_2\text{CO}_2$ towards different CRR products. This work offers new insights to the design of high-efficiency MXene-based SACs for artificial CO_2 reduction.

To maintain the stability of $\text{TM@Ti}_2\text{CO}_2$ SACs for practical application, strong binding of TM atoms with Ti_2CO_2 is a prerequisite to prevent aggregation and detachment of the TM atoms. Therefore, we first investigate the geometry and thermodynamic stability of $\text{TM@Ti}_2\text{CO}_2$. As presented in Fig. 1(a and b), two possible binding sites were considered: (i) The hollow site between three neighboring O atoms and on the top site of C atom (Site 1), (ii) the hollow site between three neighboring O atoms and on the top site of Ti atom (Site 2). As shown in the upper panel of Fig. 1(c) and

Table S1, most of the TM atoms tend to bind on the Site 1, and the value of binding energies between all the TM atoms and Ti_2CO_2 are positive, indicating a thermodynamic preference for TM atoms to be bonded to the substrate. To further investigate the kinetic stability of the TM atoms, we calculate the diffusion energy barriers of these TM atoms from the most stable site to the nearest second-stable site (Figs. S1, S2 and the lower panel of Fig. 1(c)). Previous studies demonstrated that, for the CeO_2 (1 1 1) surface, the Pd, Au and Pt atoms can rapidly diffuse with energy barriers of 0.14, 0.26, and 0.15 eV, respectively [25–27]. Given the similar surface structure of CeO_2 (1 1 1) and Ti_2CO_2 (both terminated by O atoms), we take CeO_2 (1 1 1) as a standard and select 0.3 eV as the threshold value to evaluate the kinetic stability of the TM atoms. As can be seen, the diffusion energy barriers of Cu, Pd, Ag, Pt and Au are less than 0.3 eV, indicating that they may easily diffuse on the surface and aggregate into clusters. Moreover, we find that the binding energies of these TM atoms are also relatively lower than other TM atoms, which further highlights their instability. Therefore, the Cu, Pd, Ag, Pt and Au decorated Ti_2CO_2 are considered to be unstable and excluded for further investigation.

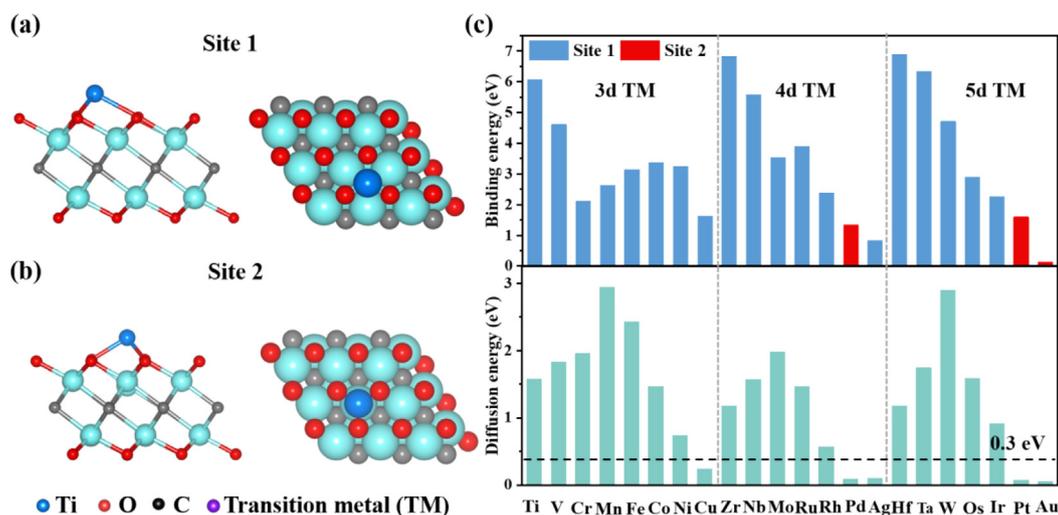


Fig. 1. The two possible adsorption sites, (a) site 1 and (b) site 2, of TM atoms on Ti_2CO_2 . (c) The binding energies and the diffusion energy barriers for different TM atoms on Ti_2CO_2 . For each SAC, only the binding energy of the most stable adsorption configuration is given, and the diffusion pathway goes from the most stable site to the nearest second stable site.

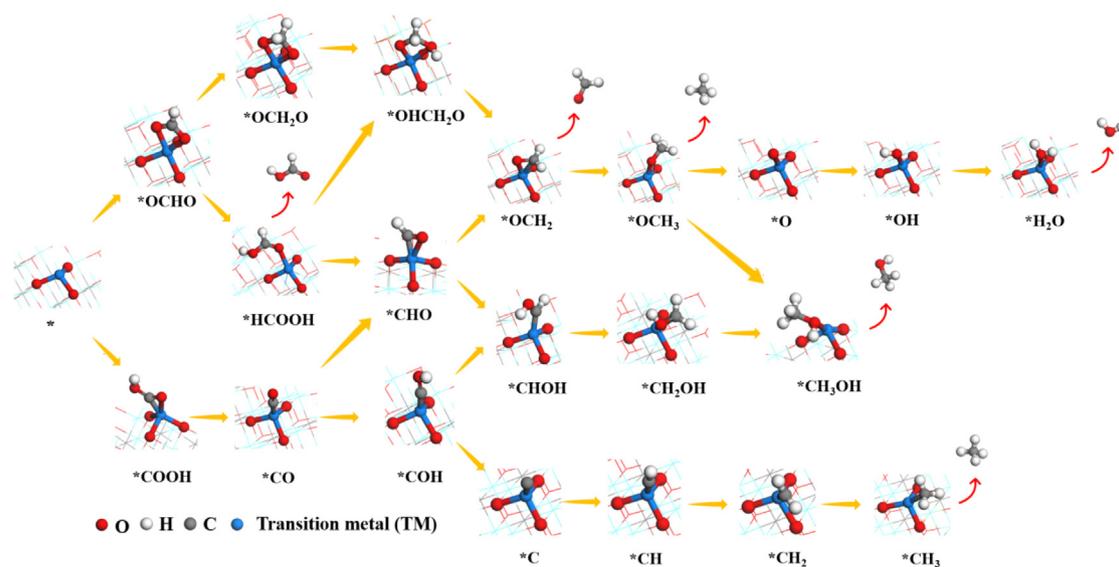


Fig. 2. Schematic of mechanisms for CO_2 reduction to different C1 products.

Using the most stable TM@Ti₂CO₂ geometries, we then examined the CRR catalytic performance of TM@Ti₂CO₂. CO₂ reduction to C1 products can proceed through different reactions that involve 2–8 electrons (Fig. 2). Particularly, due to the lack of TM ensembles in TM@Ti₂CO₂ for C–C coupling, the pathways to C2 products were not examined [28,29].

The CRR begins with the hydrogenation of CO₂ to form either carboxyl (*COOH) or formate (*HCOO) on the active center (Fig. 2), while in this process, the HER process may also consume the protons and electrons under the same reaction condition, occupying the active center and resulting in a low selectivity and Faradic efficiency. It is generally believed that the free energy change of initial protonation step for CRR (CO₂ + H⁺ + e⁻ → *HCOO/*COOH) and HER (H⁺ + e⁻ → *H) can be investigated to evaluate the selectivity [28,30]. The results are summarized in Fig. 3(a), it is found that the reaction of *HCOO formation is energetically more favorable than the *COOH formation on all TM@Ti₂CO₂ and the difference in ΔG between them is greater than 0.3 eV in most cases (see Table S2), indicating a high preference towards the *HCOO pathway on the TM@Ti₂CO₂. Therefore the *COOH reduction pathways were not further investigated and the CO product, which can only be hydrogenated from *COOH intermediate, is expected not to be formed in the TM@Ti₂CO₂ system. For all the TM@Ti₂CO₂ SACs, the calculated values of free energy changes for (H⁺ + e⁻ → *H) reaction are more positive than the first protonation step of CRR (CO₂ + H⁺ + e⁻ → *HCOO), which indicates that the TM catalytic center on Ti₂CO₂ substrate can effectively promote CRR over HER.

We further consider the protonation of *HCOO intermediate. As shown in Fig. 2, the second protons can attack the C or O atom of *HCOO, generating *HCOOH or *H₂COO intermediates. In the former case, formic acid can be formed after the direct desorption of the intermediate. The further hydrogenation of *HCOOH or *H₂COO would lead to the four-electron product of OCH₂. Nevertheless, it is unlikely to be the final product because its binding strength is relatively strong on all the TM@Ti₂CO₂ SACs (>1 eV, see Fig. S10), which indicates that *OCH₂ can serve as key intermediates for further protonation to six-electron product (CH₃OH) or eight-electron product (CH₄).

The reaction pathway for each TM@Ti₂CO₂ is determined by the most energetically favorable intermediate at each hydrogenation step. After screening all the possible intermediates, the most energetically favorable product, the potential determining step (PDS) and the reaction free energy at PDS (ΔG_{PDS}) for all TM@Ti₂CO₂

are summarized in Fig. 3(b). It is shown that TM@Ti₂CO₂ electrocatalysts can produce three kinds of C1 products, namely HCOOH, CH₃OH and CH₄, demonstrating the essential role of TM catalytic center in determining the final CRR product.

For Ti, V, Cr, Mn, Zr, Nb, Mo, Ru, Hf, Ta, W, Os and Ir SACs, CH₄ is expected to be the final product. The Mn@Ti₂CO₂ exhibits the most promising catalytic activity with the following intermediates (Fig. 3(c)): CO₂ → *HCOO → *HCOOH → *H₂COOH → *OCH₂ → *OCH₃ → *O + CH₄ → *OH → H₂O. The pathway is different from that on the Cu (1 1 1) surface and the *HCOO → *HCOOH is identified as the PDS with a ΔG_{PDS} of 0.59 eV, which is also well below that on the Cu (1 1 1) surface (0.99 eV) [31]. Intriguingly, for early transition metal SACs, the reaction pathways are uniform and the energy profiles show remarkable similarity (Fig. S4), which is probably due to their similar electronic structures. The last hydrogenation step (*OH + H⁺ + e⁻ → H₂O) is identified as the PDS with ΔG_{PDS} all greater than 1 eV (see Fig. S4), this rules out their potential to reduce CO₂ because the *OH would block the catalytic center and leads to the deactivation of the catalysts [31].

The Fe, Co and Rh SACs tend to generate CH₃OH product. Co@Ti₂CO₂ shows the highest activity, following the pathway (Fig. 3(d)): CO₂ → *HCOO → *HCOOH → *CHO → *OCH₂ → *OCH₃ → CH₃OH. The PDS is the CO₂ → *HCOO step and the corresponding ΔG_{PDS} is as low as 0.28 eV, which is well below that on Cu (2 1 1) (0.74 eV) [7] surface and competitive to other theoretically studied CO₂ to CH₃OH single atom catalysts, including V decorated β₁₂ boron monolayer (0.54 eV) [32], Ni decorated C₂N monolayer (0.31 eV) [29] and Pt supported on defective graphene (0.27 eV) [28].

Ni@Ti₂CO₂ is the only catalysts that tend to produce HCOOH with the intermediates (Fig. 3(e)): CO₂ → *HCOO → HCOOH. The PDS is identified as the CO₂ → *HCOO step with a ΔG_{PDS} at 0.29 eV, which is also significantly lower than that on Cu (2 1 1) surface (0.48 eV) [33] and is competitive to the traditional catalysts for the formation of formic acid, such as the partially oxidized atomic Co layers (0.24 V) [34] and Pd nanoparticles (NPs) dispersed on a carbon support (0.20 V) [35].

It is beneficial to search an activity descriptor to establish a “volcano” relationship on catalytic behaviors such that we can further understand the superior catalytic activity of Mn@Ti₂CO₂, Co@Ti₂CO₂ and Ni@Ti₂CO₂ SACs. It is generally agreed that the electronic structures of catalytic centers would greatly influence the electron transfer and reaction energy in catalytic processes. Hereby, we first examined the descriptor of *d* band center, which

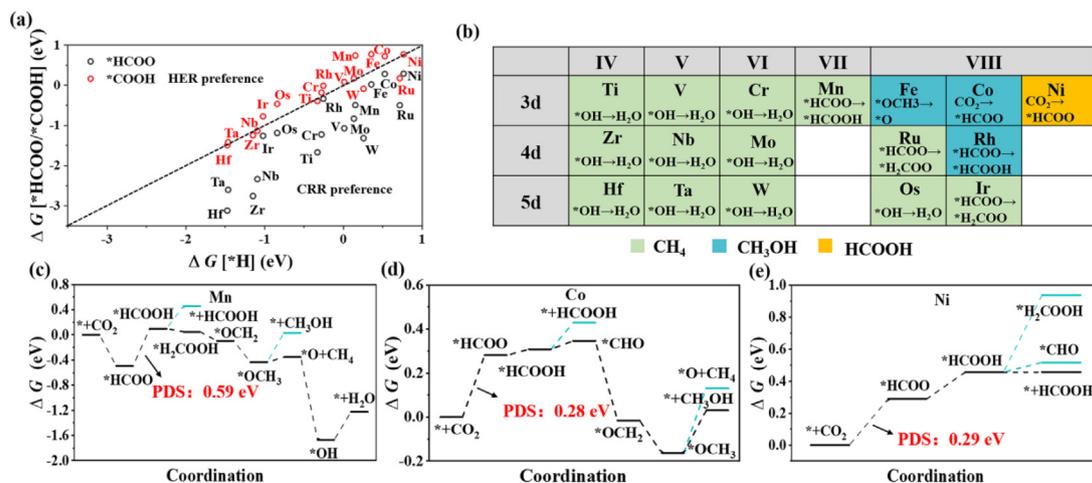


Fig. 3. (a) The reaction free energy change (ΔG) of *HCOO and *COOH formation vs HER. (b) Summary of the most energetically favorable product, the potential determining step (PDS) and the reaction free energy at PDS (ΔG_{PDS}) for each investigated TM@Ti₂CO₂. The free energy profiles for CRR catalyzed by (c) Mn@Ti₂CO₂, (d) Co@Ti₂CO₂, and (e) Ni@Ti₂CO₂.

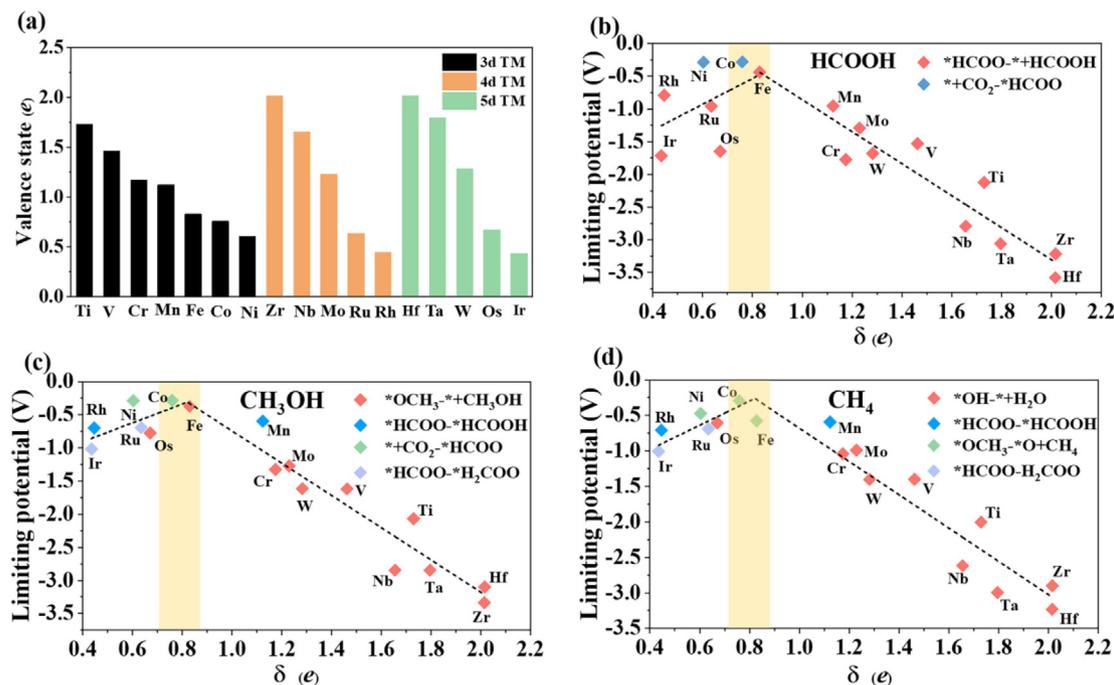


Fig. 4. (a) The valence states of TMs when they are decorated on Ti₂CO₂ substrate. The limiting potential towards (b) HCOOH, (c) CH₃OH and (d) CH₄ as a function of the valence state of TM atoms. Different colored points correspond to different PDS.

can be used to explain the general catalytic activity trend for different catalytic processes, e.g. CRR, NRR, and ORR [36,37], in bulk metals. As presented in Fig. S9, there is not obvious relationship between the *d* band center of TM@Ti₂CO₂ and the limiting potential of the three different products, which can be attributed to a different electronic structure of the single TM atoms from that in the bulk.

Based on the above results and analysis, we inspect the charge redistribution of TM atoms when it is decorated on the Ti₂CO₂ substrate through Bader charge analysis [38]. We define the valence state of different TM atoms as the number of electrons transferred from the TM atom to the O atoms. As shown in Fig. 4(a), the valence state of different TM atoms varies in a range from ~0.4 e to ~2 e, indicating that the interplay between transition metal *d* orbitals and O *p* orbitals can effectively influence the valence state of the TM atoms, and in each period, a decreasing trend can be clearly observed.

Moreover, it is known that the TM atoms can provide both empty *d* orbitals to accept electrons to form a coordination bond and lone electrons to form a covalent bond [39]. Hence, the valence state of TMs would directly influence the bonding strength with the intermediates, which is substantiated by the calculated adsorption energies of all intermediates involved in our system, as shown in Fig. S10. For all intermediates, a volcano shape relationship can be roughly observed. It can be assumed that the valence state of TMs could be used as an activity descriptor, wherefore the limiting potential towards different products as a function of the valence state of TMs is shown in Fig. 4. A volcano trend is observed for each kind of product, with the optimal scope of valence state ranging from ~0.7 e to ~0.9 e. The Fe@Ti₂CO₂ and Co@Ti₂CO₂ locate within the above region for three cases, indicating that they are the most active catalysts for CRR and tend to generate CH₃OH rather than CH₄ or HCOOH according to the reaction free energy profile. The volcano trend follows the Sabatier principle [40], which states that a promising catalyst should bind the intermediates neither too strong nor too weak. This simple descriptor well explains the catalytic performance of the TM@Ti₂CO₂ SACs, and suggest that a certain moderate degree of electron transference from TM atoms to

the substrate will be expected to promote the overall catalytic activity.

In summary, we performed DFT calculations to evaluate the potential of single TM atom decorated Ti₂CO₂ for the electrochemical reduction of CO₂ and discovered a simple descriptor that can reliably describe the catalytic performance of TM@Ti₂CO₂. Mn@Ti₂CO₂ tends to produce CH₄ with limiting potential lower than that of Cu (1 1 1) surface. Co@Ti₂CO₂ is revealed as an excellent electrocatalyst for CH₃OH production with limiting potential at -0.28 V, competitive to the most active electrocatalysts found in previous reports. Ni@Ti₂CO₂ is revealed to be a superior catalyst to produce HCOOH with limiting potential value calculated to be -0.29 V, which is well below that of Cu (2 1 1) surface. It is found that the valence state of TM atoms can effectively affect the binding strength of intermediates and is therefore proposed as a descriptor for the catalytic activity in CRR. Hereby, a volcano shaped relationship between the descriptor and the catalytic activity is demonstrated and a certain moderate degree (around 0.8 e) of TM-to-substrate electron donation would be beneficial for effective CO₂ reduction. It is expected that this simple descriptor would enrich the design principles for guiding the rational design of MXene based single atom catalysts for boarder application.

Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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Appendix A. Supplementary data

Supplementary data to this article can be found online at <https://doi.org/10.1016/j.jechem.2020.08.023>.

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